# Characterization of a high-temperature superconducting conductor on round core cables in magnetic fields up to 20 T\*

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### Abstract

The next generation of high-field magnets that will operate at magnetic fields substantially above 20 T, or at temperatures substantially above 4.2 K, requires high-temperature superconductors (HTS). Conductor on round core (CORC) cables, in which RE-Ba<sub>2</sub>Cu<sub>3</sub>O<sub>7- $\delta$ </sub> (RE = rare earth) (REBCO) coated conductors are wound in a helical fashion on a flexible core, are a practical and versatile HTS cable option for low-inductance, high-field magnets. We performed the first tests of CORC magnet cables in liquid helium in magnetic fields of up to 20 T. A record critical current  $I_c$  of 5021 A was measured at 4.2 K and 19 T. In a cable with an outer diameter of 7.5 mm, this value corresponds to an engineering current density  $J_c$  of 114 A mm<sup>-2</sup>, the highest  $J_c$  ever reported for a superconducting cable at such high magnetic fields. Additionally, the first magnet wound from an HTS cable was constructed from a 6 m-long CORC cable. The 12-turn, double-layer magnet had an inner diameter of 9 cm and was tested in a magnetic field of 20 T, at which it had an  $I_c$  of 1966 A. The cables were quenched repetitively without degradation during the measurements, demonstrating the feasibility of HTS CORC cables for use in high-field magnet applications.

(Some figures may appear in colour only in the online journal)

## 1. Introduction

Low-temperature superconductors (LTS) are limited to magnetic fields lower than 20–22 T in liquid helium due to their limited upper critical field  $B_{\rm c2}$ , which is about 27 T for Nb<sub>3</sub>Sn [1]. There is a growing need for magnets that operate at magnetic fields that exceed 20 T, or that operate at temperatures above the boiling temperature of liquid helium. These magnets include the next generation high-energy physics magnets [2], magnets for fusion experiments [3], superconducting magnetic energy storage (SMES) [4], and

other science magnets [5]. High-temperature superconductors are the only feasible materials for use in these magnets due to their very high  $B_{c2}$  values, which are expected to exceed 100 T at 4.2 K [1]. High-field magnet research is currently focused on two HTS materials: Bi<sub>2</sub>Sr<sub>2</sub>CaCu<sub>2</sub>O<sub>8</sub> (Bi-2212) wires and REBCO coated conductors. REBCO coated conductors are especially attractive candidates because of their high critical temperature of 92 K, and their high mechanical strength. Several high-field magnet programs utilizing REBCO coated conductors are underway; for example, a 32 T, all-superconducting magnet at the National High Magnetic Field Laboratory (NHMFL), and a toroid SMES at Brookhaven National Laboratory that will operate

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at a field of between 25 and 30 T [4]. These high-inductance magnets are wound from single-tape conductors that carry currents of no more than a few hundred amperes at their operating field and at 4.2 K. The single-tape windings are subject to stringent conductor requirements with respect to conductor homogeneity to prevent local hotspots in the magnet. The high magnet inductance makes quench protection very challenging because of the very high voltages that could develop during a quench.

Fast-ramping magnets for high-energy physics and fusion applications require a low inductance and therefore need high-current windings. Accelerator magnets that are currently operated at the large Hadron collider (LHC) at CERN operate at a winding current of 11 850 A and a J<sub>e</sub> of about  $300-400 \text{ A mm}^{-2}$  at 8.3 T. The central solenoid in the international thermal-nuclear experimental reactor (ITER) is designed to operate at a winding current of 46000 A and a winding current density  $J_{\rm w}$  of about 19 A mm<sup>-2</sup> at 13 T [6].  $J_{\rm w}$  is the current density in the overall magnet windings, taking structural materials and voids between the round windings into account. One of the main challenges for HTS to be utilized in low-inductance magnet applications is for them to be bundled into high-current cables. To be feasible for high-field magnet applications, these cables need to be relatively flexible, have a high cable current of at least several kiloamperes, and a  $J_e$  of at least 50 A mm<sup>-2</sup> at the operating field and temperature.

Several HTS cabling methods are being explored; for example, Bi-2212 wires have been cabled into a Rutherford configuration [7] and a cable current of 1700 A at 15 T has been predicted [8]. There are still major challenges that need to be overcome with Bi-2212 wires before they become a viable option for high-field magnets. These challenges include raising the critical current density  $J_c$ , optimizing the conductor processing conditions to reduce leakage and bubble formation [9], and significantly raising their mechanical strength [10].

Three approaches are being pursued to cable REBCO coated conductors into a configuration applicable to high-field magnets. Patterned REBCO coated conductors are assembled into a Roebel bar configuration [11] with a demonstrated critical current at 77 K of several kiloamperes in selffield [12]. The Twisted Stack approach uses multiple REBCO coated conductors stacked into a rectangular configuration and then twisted [13]. A critical current of 1530 A in self-field has been demonstrated in this cable configuration at 77 K [14]. These two cable configurations have the disadvantage that the cables are not very flexible and that current sharing between individual tapes in the cable is limited. Initial data on the in-field performance of these cables has been presented, but no data have yet been published. Additionally, longitudinal defects are frequently present in REBCO coated conductors [15, 16], which severely limit the performance of Roebel cables because the patterning of the strands forces the current to cross these defects.

Advanced Conductor Technologies LLC is currently developing CORC cables for power transmission and high-field magnet applications [17]. We have demonstrated

that CORC cables, in which multiple coated conductors are wound in a helical fashion on a round former, are relatively flexible and can potentially be wound into magnets that require tight cable bends [18, 19]. The high level of elasticity of REBCO coated conductors, especially under axial compressive strain [20], enables them to be wound on relatively small formers of only several millimeters in diameter without experiencing irreversible degradation. Additionally, the anisotropic nature of the reversible strain effect in REBCO coated conductors [16], in combination with the conductor winding angle of between 30° and 60° in CORC cables, results in only a minimal reversible reduction of the conductor performance due to both winding and operating strains. Several high-current CORC cables have been demonstrated, the most prominent cable having an  $I_c$ of 7651 A at 76 K in a cable with outer diameter of only 10 mm [21].

Here, we present the results of the first measurements of CORC cables at high magnetic field up to 20 T, which were performed at the user facility of the NHMFL. An insert to provide the CORC cables with the mechanical support needed to withstand the Lorentz forces generated by multi-kiloampere currents in high magnetic fields was developed for the 20 cm bore resistive magnet at the NHMFL. Five CORC cables, one wound into a small magnet, were successfully tested and demonstrate the feasibility of high- $J_{\rm e}$  CORC cables for low-inductance magnet applications.

# 2. Experimental details

### 2.1. Conductor on round core cables

A total of five CORC cable samples were prepared for testing in a background magnetic field of up to 20 T. An overview of the different cables, including information on the former material and the outer diameter of the cables, the number of tapes and layers in the cables, and the details of the tapes with which the cables were wound, is provided in table 1. The cables were assembled from 4 mm wide REBCO coated conductors from SuperPower, Inc., consisting of a 50  $\mu$ m thick Hastelloy C-276 substrate, a 1  $\mu$ m thick superconducting layer, and a 20  $\mu$ m thick surround plated copper layer. Some of the cables were constructed from coated conductors that contained advanced pinning centers through nanoparticles and are designated AP for 'advanced pinning'. Coated conductors that did not contain additional pinning centers are designed ST for 'standard'.

Cable CORC-1 was constructed from a 5.0 mm diameter solid aluminum former and 6 ST coated conductors. The former was pre-shaped into a 11 cm diameter loop to fit the cable support structure (see section 2.2) and to ensure minimum cable degradation due to cable bending. Additional mechanical support was provided by filling the cable with 51 wt%In-32.5 wt%Bi-16.5 wt%Sn solder after winding. The cable was insulated with a layer of polyamide tape.

Cable CORC-2 was prepared by use of a flexible, stranded-copper, insulated former that had an outer diameter of 5.5 mm. The 1.2 m-long, straight cable consisted of 12 ST

<b>Table 1.</b> CORC cable spe	CITICATIONS.
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Sample #	Former/diameter (mm)	# of tapes/# of layers	Current capacity	Conductor type/ <i>I</i> <sub>c</sub> (76 K, 0 T) (A)	Cable outer diameter (mm)
CORC-1	Solid Al/5.0	6/2	Low	ST/120	5.5
CORC-2	Insulated stranded Cu/5.5	12/4	Low	ST/120	6.5
CORC-3	Insulated stranded Cu/4.0	40/13	High	9 ST, 31 AP/130, 110	7.5
CORC-4	Insulated stranded Cu/3.2	52/17	High	AP/120	7.5
CORC-5	Insulated stranded Cu/5.5	20/6	High	10 ST, 10 AP/130, 110	7.0



**Figure 1.** Cable CORC-2 after it was bent into a 12 cm diameter loop.

tapes that were wound into 4 layers, and an outer layer of polyamide insulation. The cable was bent into the 11 cm diameter loop after it was wound (see figure 1).

Cable CORC-3 was constructed by use of a flexible, insulated, stranded-copper former with a diameter of only 4.0 mm. The initial 4 layers in the cable were wound from 9 ST coated conductors, on top of which an additional 31 AP coated conductors were wound into 8 layers. The cable consisted of a total of 40 coated conductors in 13 layers and had an outer diameter of 7.5 mm. Cable CORC-4 contained 52 AP conductors that were wound into 17 layers onto a 3.2 mm diameter former. The cable had an outer diameter of 7.5 mm.

Cable CORC-5 was a cable in which 10 AP and 10 ST coated conductors were wound into 6 layers onto a flexible, insulated, stranded, copper former with a diameter of 5.5 mm. The 6 m-long cable was wound onto an aluminum cylinder, forming a two-layer, 12-turn magnet. The magnet was 4.5 cm in height, had an inner diameter of 9 cm and an outer diameter of 12 cm. Figure 2 shows the inner 6 turns of the magnet after they were wound onto the aluminum cylinder. The magnet was filled with epoxy after it was inserted into the support structure.

# 2.2. Cable support structure

Cables CORC-1 to CORC-4 were wound into a 1.5-turn loop with a 11 cm diameter (see figure 1) to ensure that the orientation of the external magnetic field was normal to the cable. A structure was developed to support the CORC cables against Lorentz forces of up to 40 kN per turn at 19 T



**Figure 2.** The 6 turns of the inner layer of the magnet that was wound from sample CORC-5.

and a sample current of 6000 A. The 1.5-turn cable loops were inserted into an aluminum cup located at the bottom of the support structure, which was then filled with epoxy (see figure 3(a)). The cable sections close to the terminals were supported by a G-10 cable spacer (see figure 3(a)). The cable, with its support structure, was bolted to the inside of an aluminum tube (figure 3(b)) located at the tail end of the insert structure (figure 3(c)) and was connected to the vapor-cooled current leads that were rated at 7500 A.

### 3. Results

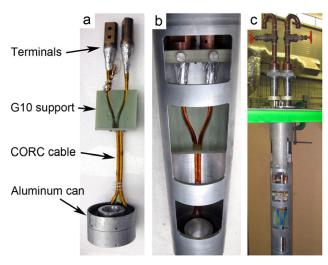
Several CORC cables were studied in a background field as high as 19.8 T. These cables included two low-current CORC cables that had an  $I_{\rm c}$  of less than 1000 A at 4.2 K and 19.8 T, and three high-current CORC cables that had an  $I_{\rm c}$  far exceeding 1000 A under these conditions. Table 2 lists the performance of the five HTS CORC cables together with the performance of LTS cables used in some of the main low-inductance magnets that are either in operation or are being developed.

# 3.1. Low-current CORC cable tests

Measurements of cable CORC-1 at 76 K in self-field were conducted at NIST before the cable was tested at the NHMFL. Throughout this paper, the critical current is defined at an electric field criterion of 1  $\mu$ V cm<sup>-1</sup>, with the distance

Cable	<i>I</i> <sub>c</sub> (19.8 T) (A)	<i>I</i> <sub>c</sub> (12 T) (A)	J <sub>e</sub> (19.8 T) (A mm <sup>-2</sup> )	J <sub>e</sub> (12 T) (A mm <sup>-2</sup> )	$J_{\rm w}~(19.8~{\rm T})~({\rm A~mm}^{-2})$	J <sub>w</sub> (12 T) (A mm <sup>-2</sup> )			
CORC-1	526	686	22	29	17	23			
CORC-2	877	1120	26	33	21	27			
CORC-3	3327	4422	76	101	60	80			
CORC-4	5021 (19 T)	_	114 (19 T)	_	89 (19 T)				
CORC-5	1966	2657	51	69	40	54			
ITER central solenoid	_	46 000 (13 T)	_		_	19 (13 T)			
(under development)		•							
NHMFL 45 T hybrid	_	10 000 (15.3 T)	_	50 (15.3 T)	_	40 (15.3 T)			
LTS outsert [22]									

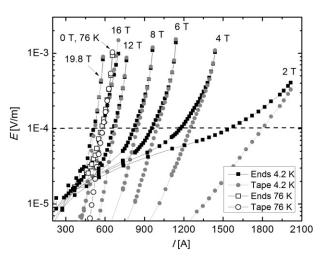
Table 2. Performance of HTS CORC cables and LTS magnet cables.



**Figure 3.** (a) CORC cable with local reinforcement provided by the epoxy-filled aluminum cup and the G-10 structure near the cable ends. (b) Cable inserted into the lower part of the support structure. (c) Magnet insert containing the cable sample and vapor-cooled current leads.

measured along the tapes in the cable, unless stated otherwise. The n-value, which is an indication of the steepness of the superconducting transition at an electric field of 1 mV cm<sup>-1</sup>, was 13.5. The critical current was determined by means of a set of voltage contacts that were located on the copper terminals of the cable and subtracting the resistive voltage from the overall voltage. This resulted in an  $I_c$  of 593 A and a total contact resistance between the copper cable terminals and the superconducting tapes of 75 n $\Omega$  at 76 K. The method for determining  $I_c$  from the voltage across the cable terminals is used throughout this paper unless stated otherwise. In some cases, a voltage tap across a single tape is used to determine the cable  $I_c$ , but this method relies on a uniform current distribution, and an error will occur if the joint resistance, and therefore current distribution, is not equal for all tapes. All measurements of the cables were performed up to a current at which the cables quenched. The cables were protected with a quench protection circuit that caused the sample current to cut off when the voltage at the cable terminals exceeded 2 mV.

The cable  $I_c$  of cable CORC-1 was measured as a function of applied magnetic field at 4.2 K in the resistive magnet at the NHMFL (it was 526 A at 19.8 T). The joint resistance of sample CORC-1 at 4.2 K was 25 n $\Omega$  for both joints combined.

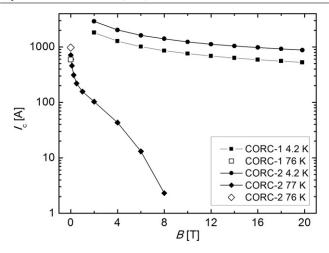


**Figure 4.** E-I characteristics of cable CORC-1, measured with two sets of voltage contacts as a function of magnetic field at 4.2 K. One set was located on the copper terminal (squares) and one set was located on one of the tapes in the outer layer of the cable (circles). Included in the figure are the E-I curves that were measured at 76 K in self-field (open symbols). The dashed line indicates the electric field criterion of 1  $\mu$ V cm<sup>-1</sup>.

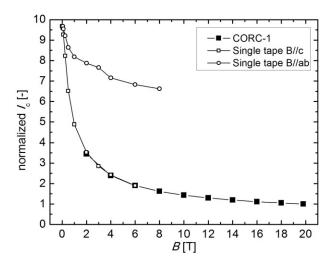
Since the magnetic field was only applied to the 1.5 turns, the effective tape length between the voltage contacts was reduced from about 1.5 m in self-field to only 0.4 m with applied field.

The electric field versus current (*E-I*) characteristics of cable CORC-1, measured with two sets of voltage contacts, are shown in figure 4 for different magnetic fields at 4.2 K. The *E-I* curves that were measured at 76 K in self-field are also included in the figure and are indicated by open symbols. The critical current at 76 K in self-field corresponds to that at about 16 T and 4.2 K. The resistive transition at low currents from which the contact resistance between cable terminals and tapes is determined is clearly visible in the *E-I* characteristics that were measured with the voltage contacts on the terminals (squares). Such a resistive transition is not measured with the voltage contacts that were located on one of the tapes in the outer layer of the cable (circles).

The magnetic field dependence of  $I_c$  at 4.2 K of cable CORC-1, determined with the voltage contacts on the cable ends, is shown in figure 5, together with  $I_c$  measured at 76 K in self-field. Unfortunately, the maximum current of the power supplies available at the time was 2000 A, which



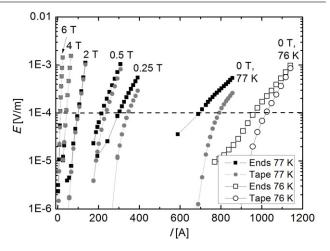
**Figure 5.** Magnetic field dependence of  $I_c$  for cable CORC-1 at 4.2 K and at 76 K, and of cable CORC-2 at 4.2, 76 and 77 K.



**Figure 6.** Magnetic field dependence of the normalized  $I_c$  at 4.2 K of cable CORC-1 and of a superconducting tape from a similar batch of ST coated conductors. The magnetic field dependence of  $I_c$  of the single tape is plotted for magnetic fields applied parallel to the c-axis and parallel to the ab-plane of the REBCO layer. The critical current of cable CORC-1 is normalized to its value at 19.8 T, while the tape  $I_c$  for both field orientations is multiplied by a factor of 0.014, such that its  $I_c$  at 6 T for field applied parallel to the c-axis corresponds to that of the cable at 6 T.

prevented us from measuring  $I_{\rm c}$  at magnetic fields below 2 T. The self-field generated by the cable has not been taken into account, because it is much less than the applied magnetic field.

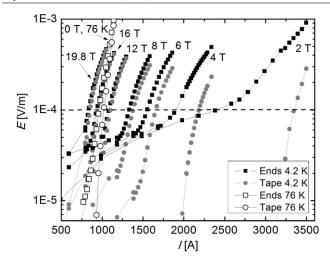
The magnetic field dependence for  $I_c$  of cable CORC-1 at 4.2 K for fields up to 19.8 T is compared in figure 6 to that of a single coated conductor from a similar batch for fields up to 8 T. Both conductors in the cable and the individual coated conductor were from the standard type, which did not included advanced pinning centers. The magnetic field dependence of  $I_c$  for the single coated conductor was measured for magnetic fields applied parallel to the c-axis and for magnetic fields applied parallel to the ab-planes of the REBCO layer. The critical current of the



**Figure 7.** *E-I* characteristics of cable CORC-2, as measured with two sets of voltage contacts. One set was located on the copper terminal (squares) and one set was located on one of the tapes in the outer layer of the cable (circles). All data were measured at 77 K, except for the two curves that were measured at 76 K in self-field (open symbols). The dashed line indicates the electric field criterion of 1  $\mu$ V cm<sup>-1</sup>.

cable is normalized to its value at 19.8 T. For comparison, the  $I_c$  of the tape for field parallel to the c-axis (=136 A) is scaled to match the cable  $I_c$  at 6 T, requiring multiplication by a factor of 0.014. The normalized  $I_c$  for the cable coincides with that of the tape with field applied parallel to the c-axis over the range of 2–6 T, indicating that the performance of CORC cables is determined by the magnetic field component parallel to the c-axis of its tapes. This is expected because the magnetic field orientation experienced by each tape in a CORC cable varies over its length between the case of parallel to the c-axis and that of parallel to the ab-plane. The locations in each tape corresponding the lowest  $I_c$ , which are those where the magnetic field is oriented along the c-axis, will determine the overall dissipation in each tape in the cable. Comparison of the cable  $I_c$  with the tape  $I_c$  in magnetic fields applied parallel to the ab-plane shows that the cable  $I_c$  at high field can potentially be raised by a factor of 4 through improvements of the flux pinning in these conductors that contain no enhanced pinning centers.

The *E–I* characteristics of cable CORC-2 were measured at NIST for 76 K in self-field before the cable was wound into a 1.5-turn loop. The same characteristics were measured as a function of magnetic field at 77 K at the NHMFL after the cable was wound into the loop (see figure 7). The voltages were measured with two sets of voltage contacts; one set was located on the copper cable terminals (squares) and one set was located on one of the tapes in the outer layer of the cable (circles). The overall contact resistance was about 25 n $\Omega$  at 76 K. The  $I_c$  that was measured at the NHMFL at 77 K in self-field after the cable was wound into the 1.5-turn loop was only 714 A, compared to 976 A at 76 K before it was wound into the loop. This large difference indicates that some damage to the cable occurred during winding because the expected critical current at 77 K is about 850 A, assuming a reduction in  $I_c$  of about 15% when the temperature is raised from 76 to 77 K.

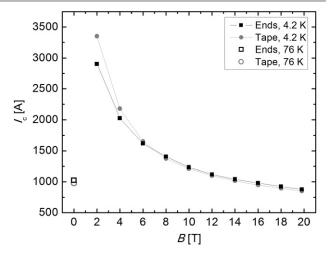


**Figure 8.** E-I characteristics of cable CORC-2 as a function of magnetic field at 4.2 K, determined with two sets of voltage contacts; one set was located on the copper terminal (squares) and one set was located on one of the tapes in the outer layer of the cable (circles). Data taken at 76 K in self-field are included in the figure (open symbols). The dashed line indicates the electric field criterion of 1  $\mu$ V cm<sup>-1</sup>.

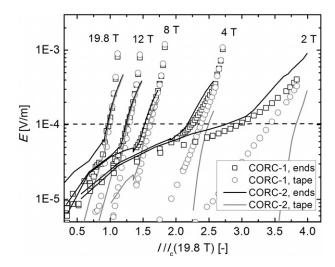
The E-I characteristics of cable CORC-2, that were measured as a function of magnetic field at 4.2 K, are shown in figure 8. The critical current of the cable was 877 A at 19.8 T, which corresponds to a  $J_{\rm e}$  of 26 A mm<sup>-2</sup> (see table 2). The overall contact resistance was about 15 n $\Omega$  at 4.2 K. The critical current could not be determined for fields below 2 T because of the 3500 A current limit of the power supplies during the test of sample CORC-2. The magnetic field dependences of  $I_{\rm c}$  for cable CORC-2 at 4.2 K and at 77 K are compared to those of cable CORC-1 in figure 5.

# 3.2. Current distribution effects

The minor damage that occurred in cable CORC-2 when it was wound into the 11 cm, 1.5-turn loop resulted in an inhomogeneous current distribution between its tapes that is most pronounced at magnetic fields below 8 T. The inhomogeneous current distribution is evident from the E-I characteristics that are shown in figure 8. The superconducting transition determined with the voltage contacts on the cable ends occurs at a much lower cable current compared with the transition, when measured on one of the tapes in the outer layer of the cable. Once it occurs, the superconducting transition of the particular tape occurs more rapidly than that of the overall cable, which is indicated by a relatively high *n*-value. The *n*-value of the superconducting tape increases from about 10 at high fields, where the current distribution is more homogeneous, to 23 at fields below 8 T. The n-value of the cable that is determined with the voltage contacts on the terminals is in the range of 12–18 independent of magnetic field. The magnetic field dependence of  $I_c$  of cable CORC-2 that is shown in figure 9 also reveals the relatively large difference in I<sub>c</sub> at fields below 8 T when determined with two different sets of voltage contacts. These measurements show



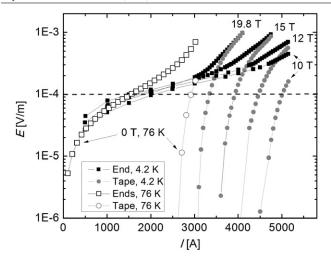
**Figure 9.** Magnetic field dependence of  $I_c$  of cable CORC-2 as determined with two sets of voltage contacts. Included in the figure are the critical currents of the cable at 76 K in self-field (open symbols).



**Figure 10.** Electric field as a function of normalized current of cables CORC-1 and CORC-2 at different magnetic fields at 4.2 K. The applied currents are normalized to their cable  $I_c$  values at an applied field of 19.8 T to allow a direct comparison between the two cables that have different overall critical currents. The dashed line indicates the electric field criterion of 1  $\mu$ V cm<sup>-1</sup>.

that the most reliable method to determine the overall cable  $I_c$  with a single tap is with the voltage contacts on the terminals, and not with one located on only one tape.

The inhomogeneous current distribution in cable CORC-2 at magnetic fields below 8 T is also evident when the *E-I* characteristics of the cable are compared with those from cable CORC-1, in which the current distribution is more homogeneous. The electric field of both cables, as measured with the voltage contacts on the copper terminals and on one of the outer tapes, is plotted in figure 10 as a function of normalized current for different magnetic fields. The electric fields that were measured with the voltage contacts on the cable terminals are comparable for both cables at all magnetic fields. On the other hand, the electric fields that were measured on one of the tapes in the outer layer of the



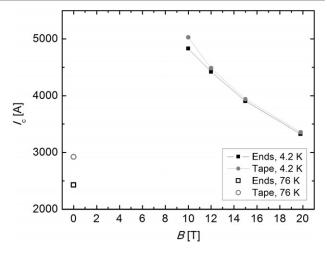
**Figure 11.** *E–I* characteristics for cable CORC-3 at 76 K in self-field and at 4.2 K for magnetic fields ranging from 10 to 19.8 T. One set of voltage contacts was located on the copper terminal (squares) and one set was located on one of the tapes in the outer layer of the cable (circles). The dashed line indicates the electric field criterion of 1  $\mu$ V cm<sup>-1</sup>.

cables show a large deviation between the two cables at fields below 8 T. The overall cable electric field in cable CORC-2 reaches 1  $\mu$ V cm<sup>-1</sup> at a cable current of 2901 A at 2 T, compared to 3350 A when the transition is measured on one of the tapes. Such a large deviation in the current, for which the superconducting transition occurs, was not measured in cable CORC-1, even at 2 T, nor was it measured in sample CORC-2 at 76 K in self-field, before being wound into the loop (see figure 8). Therefore, we conclude that the current distribution in cable CORC-2 is inhomogeneous and that the inhomogeneous distribution is most likely caused by damage that occurred during winding of the cable into the 11 cm diameter loop.

## 3.3. High-current CORC cable tests

Cable CORC-3 was wound from 40 coated conductors to demonstrate the feasibility of a high cable  $I_{\rm c}$  at high magnetic fields. A cable  $I_{\rm c}$  of 2426 A was measured at 76 K after the cable was wound into a loop of 11 cm diameter. A critical current of 3327 A was measured at 4.2 K and 19.8 T, which corresponds to a  $J_{\rm e}$  of 76 A mm<sup>-2</sup> and a  $J_{\rm w}$  of 60 A mm<sup>-2</sup>. The cable current was raised to a quench current of 4101 A at 19.8 T. The E-I characteristics of cable CORC-3 are shown in figure 11 for magnetic fields between 10 and 19.8 T at 4.2 K, and at 76 K in self-field. They were measured with two pairs of voltage contacts, one pair located on the cable terminals (squares) and one pair located on one of the tapes in the outer layer (circles). The total contact resistance between the cable terminals and the tapes was about 60 n $\Omega$  at 76 K and about 25 n $\Omega$  at 4.2 K.

The critical current of cable CORC-3 was measured for fields as low as 10 T, at which field the sample current exceeded 5000 A, the limit of our setup at the time. Figure 12 shows the magnet field dependence of  $I_c$  as determined with



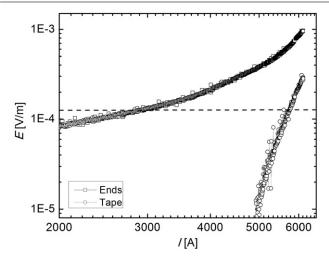
**Figure 12.** Magnetic field dependence of  $I_c$  of cable CORC-3 at 4.2 K, determined with two sets of voltage contacts. One set was located on the copper terminal (squares) and one set was located on one of the tapes in the outer layer of the cable (circles). The cable  $I_c$ s at 76 K in self-field are included (open symbols).

both sets of voltage contacts. The cable  $I_c$  at 76 K in self-field is also included in the figure. The AP coated conductors used in part to construct cable CORC-3 resulted in a much higher in-field performance at 4.2 K, compared to that of cables CORC-1 and CORC-2, which contained only ST coated conductors. As a result, the  $I_c$  of cable CORC-3 at 76 K in self-field corresponds to  $I_c$  at 4.2 K for a field exceeding 19.8 T, compared to about 16 T for sample CORC-1 and about 15 T for cable CORC-2. The advanced pinning in most of the conductors of cable CORC-3 has resulted in an enhancement of  $I_c$  at 19.8 T of at least 50%, compared to the standard conductors used in cables CORC-1 and CORC-2, when compared to their  $I_c$ s at 76 K in self-field.

Cable CORC-4 was tested at 4.2 K in a background field of 19 T. The E-I characteristics for the measurement are shown in figure 13. The cable had a record critical current of 5021 A, or a  $J_e$  of 114 A mm<sup>-2</sup> at 19 T, as determined from the voltage contacts on the cable terminals. The contact resistance between the superconducting tapes and both cable terminals was 23 n $\Omega$ . The maximum sample current was 6116 A, at which the sample quenched. The critical current could not be measured at lower magnetic fields, or at 76 K, because of the current limit of our setup at the time.

# 3.4. Magnet wound from CORC cable

The small magnet that was wound from cable CORC-5 was tested at 4.2 K in background fields ranging from 12 to 19.8 T. This magnet is the first HTS cable magnet tested in a background field of this magnitude. The *E-I* characteristics of the coil measured in a background field of 12 T, with four different sets of voltage contacts, are shown in figure 14(a). The critical current determined with the contacts on the terminals was 2657 A at 12 T. The voltage drop at a cable current of about 1000 A, as measured with three voltage pairs, indicates a redistribution of the current within the cable.

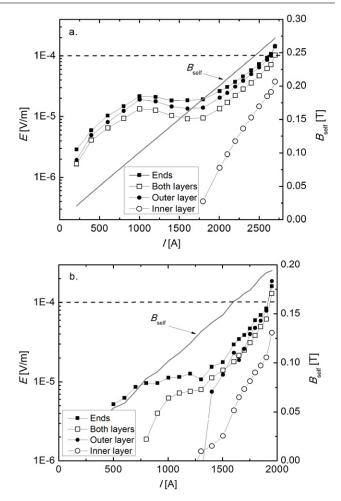


**Figure 13.** E-I characteristics for cable CORC-4 at 4.2 K and a magnetic field of 19 T. One set of voltage contacts was located on the copper terminal (squares) and one set was located on one of the tapes in the outer layer of the cable (circles). The dashed line indicates the electric field criterion of 1  $\mu$ V cm<sup>-1</sup>.

The voltage contact pair located over the inner layer of the coil (open circles) does not measure such a voltage drop. The current redistribution is most likely the result of damage near the end of the top winding in the outer layer of the magnet, that occurred during winding, where a small kink in some of the tapes in the cable was visible. The three pairs of voltage contacts, that show the voltage redistribution signature, are all located over the damaged cable section. The current redistribution is less apparent in a background field of 19.8 T (see figure 14(b)). The magnetic field that was generated by the magnet  $B_{self}$  was measured at the magnet axis with a Hall probe and was 0.2 T at a cable current of about 2000 A, resulting in a total field of 20 T. The magnetic field of the magnet, as a function of current, is included in figures 14(a) and (b), indicating a linear dependence with a slope of  $0.1 \text{ mT A}^{-1}$ . The critical current of the magnet at a total field of 20 T was 1966 A, while the Ics of the inner and outer layers were 2014 A and 1916 A, respectively. The magnet  $I_c$  of 1966 A corresponds to a  $J_e$  of 51 A mm<sup>-2</sup> and a  $J_{\rm w}$  of 40 A mm<sup>-2</sup> at 4.2 K and 20 T.

The critical current of the total magnet and its inner and outer layers was measured at different magnetic fields as low as 12 T (see figure 15). The overall  $I_c$  in the magnet is determined mainly by the outer layer that contains the damaged section. At the time, the current supplies available were limited to a maximum current of 3500 A, which was insufficient to determine  $I_c$  at fields below 12 T. A maximum current of 3500 A was applied to sample CORC-5 at 4.2 K in self-field, during which it generated a field of 350 mT at the magnet axis.

A comparison with LTS cables that operate at fields between 13 and 15.3 T shows that cable CORC-3 already exceeds their performances with respect to  $J_{\rm W}$  (see table 2). These magnets could potentially operate at a temperature of about 20 K when constructed from a CORC cable. For instance,  $J_{\rm W}$  of 89 A mm<sup>-2</sup> at 19 T and 4.2 K in CORC-3



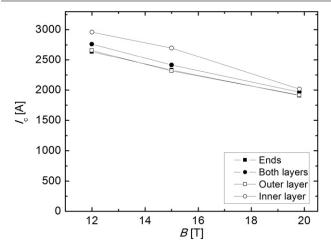
**Figure 14.** (a) E-I characteristics of the magnet wound from cable CORC-5 at 4.2 K in a background field of 12 T. (b) E-I characteristics at 4.2 K in a background field of 19.8 T. Four sets of voltage contacts were used in the measurements, as indicated in the figure. Included in both figures is the measured magnetic field generated by the coil. The dashed line indicates the electric field criterion of 1  $\mu$ V cm<sup>-1</sup>.

is expected to correlate to about 74 A mm<sup>-2</sup> at 12 T and 20 K [23], which greatly exceeds  $J_{\rm w}$  of the ITER Central Solenoid. Additionally, the performance of CORC cables at 20 T and 4.2 K could potentially be raised towards a  $J_{\rm e}$  of 200–600 A mm<sup>-2</sup> by reducing the thickness of the substrate of the tapes and by raising the conductor in-field  $I_{\rm c}$  through an increase in thickness of the REBCO layer and through further improvement of its high-field pinning performance.

The measurements demonstrate the feasibility of operating CORC cables at currents that exceed 5000 A in magnetic fields of 19 T. The results also show a clear path to further enhance the in-field performance of CORC cables towards very high  $J_{\rm e}$ s that are needed in, for instance, accelerator magnets.

### 4. Conclusions

We successfully performed the first measurements of hightemperature superconducting cables in background fields as



**Figure 15.** Magnetic field dependence of  $I_{\rm c}$  of the magnet wound from cable CORC-5 at 4.2 K for both individual magnet layers, and for both layers combined.

high as 20 T. Several conductor on round core cables were tested, demonstrating a critical current as high as 5021 A at 19 T and 4.2 K, which corresponds to an engineering current density of 114 A mm<sup>-2</sup>. We performed the first measurements of a magnet wound from HTS cables at a total magnetic field of 20 T, for which it had a critical current of almost 2000 A was demonstrated. The results demonstrate the feasibility of HTS CORC cables for low-inductance, high-field magnet applications. A comparison shows that electrical performance of some of the HTS CORC cables already exceed the performance of existing LTS cables that are currently being operated or developed. Magnets such as the ITER Central Solenoid, or the LTS outsert of the 45 T hybrid magnet at the NHMFL, could potentially be operated at higher magnetic fields, or at temperatures exceeding 20 K when constructed from HTS CORC cables.

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to foster better understanding. Such identification implies neither recommendation nor endorsement by NIST, nor that the materials identified are necessarily the best available for the purpose.

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